## POLYMERS WITH SILICON-CARBON CHAINS FRAMED BY SILACYCLOALKANE GROUPS

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We have synthesized spiran compounds with the silicon-carbon hetero- $\overline{\text{cycles}}$   $\overline{\text{CH}_2(\text{CH}_2)_{n-1}\text{Si}(\text{CH}_2)_2\text{CH}_2(n=4, 5)}$  and their analogs with alkyl radicals  $RR'SI(CH_2)_2CH_2$   $(R=C_2H_5; R'=n-C_3H_7; C_2H_8)$ , and also the tricyclic compound  $\frac{|CH_2(CH_2)_3Si|CH_2}{|CH_2/CH_2|CH_2}$ . Their heat-induced polymerization and the properties of the resulting polymers have been investigated, The influence of the groups  $I_{\vert}$ -CH<sub>2</sub>(CH<sub>2</sub>) <sub>n-1</sub>Si= (n=4,5) framing the main chain of the polymers on the stability of the chain to thermal oxidation and on its flexibility has been studied,

The present work is a continuation of preceding investigations on the synthesis of polymers with silicon-carbon main chains and on the study of their properties [1-5]. By the polymerization of strained siliconcarbon rings we have previously obtained high-molecular-weight compounds of the type of  $[-R_2S_{i}-R'-]_n$ , where R' is an alkylene or arylene group and R is alkyl, aryl, or alkenyl.

We have effected the synthesis of polymers with a silaeycloalkane framing of the main chain of the type

 $[-CH_2(CH_2)_k\dot{SL}(CH_2)_{m}$ -]<sub>n</sub>, where m = 1 or 3 and k = = 3 or 4 and have given an evaluation of the contribution made by the silacycloalkane framing to the properties of the silicon-carbon polymer. The introduction into a polymer chain of silacycloalkane, and particularly silacyclohexane, rings is of interest for a number of reasons. Thus, it is known from the case of individual compounds of the type  $R_2Si(CH_2)_4CH_2$  that the silacyclo-J **<sup>I</sup>** hexane group is stable to the action of strong acids and bases and possesses an increased thermal stability  $[6-8]$ . The endocyclic Si-C bond of the silacyclohexane group is extremely stable to the action of electron impact [9] and, finally, 1, 1-di(n-alkyl)silacyelohexanes

possess low glass points (from  $-50$  to  $-60$ °C) [10]. The monomerie compounds used for the synthesis of these polymers were obtained by the following routes:



The reaction was carried out as described previously [11, 12]. The properties and yields of the substances obtained are given in Table 1. Their purity

was checked by means of gas-liquid chromatography and was  $\sim 96\%$  for I and  $\sim 99.5\%$  for II and III.

In addition, we have obtained  $1, 1$ -diethyl-1-silacyclobutane  $\text{CH}_2(\text{CH}_2)_2\text{Si}(C_2\text{H}_5)_2$  (IV) and 1-ethyl-1-ni propyl-1-silacyclobutane CH<sub>2</sub>(CH<sub>2</sub>)<sub>2</sub>Si(C<sub>2</sub>H<sub>5</sub>)CH<sub>2</sub>CH<sub>2</sub>CH<sub>3</sub>

(V), and these have been used for the preparation of polymers the properties of which have been compared with the properties of polymers of derivatives of the spiran compounds.

Compound IV was synthesized from  $\rm CH_2(CH_2)_2SiCl_2$  $1-\frac{1}{2}$ and  $C_2H_5MgBr$ , and V from  $CH_2(CH_2)_2Si(C_2H_5)Cl$ and  $C_3H_7MgBr$ . Their properties are also given in Table 1. Compound IV is an open-chain analog of II, and compound V an analog of III.

Thermal polymerization of the spiran compounds was carried out as a block process by the method used for silacyclobutanes [2]. Immediately before polymerization, the monomers were redistilled and were dried over lithium hydride or calcium hydride, and were then freed from dissolved gases in a vacuum of  $\sim 10^{-2}$  mm. The polymer blocks were dissolved in benzene and precipitated from the solution with methanol, after which the polymer was brought to constant weight at 80-100° C  $(10 \text{ mm})$ .

Polymerication of the dialkylsilacyclobutanes IV and V was carried out in purified n-decane (1 mole of monomer 4-6 moles of solvent). The temperature of polymerization, the yields of the polymers, and their characteristic viscosities, elementary compositions, and molecular weights are given in Table 2. In the block polymerization of IV and V, weakly cross-linked polymeric products are formed which swell readily in the usual organic solvents, This difference in the polymerization of the silacyclobutanes containing ethyl and propyl groups from the spiran compounds II and III and from 1,1-dimethyl-l-silacyclobutane [2] is due, in our opinion, to the reduced resistance of the  $Si-Calkyl$  bond to homolytie decomposition. In actual fact, according to the literature  $[15-17]$ , the energy of the Si- $CH<sub>3</sub>$  bond is 75 kcal/mole, that of the  $Si-C<sub>2</sub>H<sub>5</sub>$  bond 62 kcal/mole, and that of the  $Si-C_3H_7-n$  bond 57 kcal/mole. At the same time, the endocyclic Si-C bond in the silacyclopentanes or the silacyclohexanes is even more stable than the  $Si-CH_3$  bond [9]. Consequently it is likely that in bulk polymerization (exothermic reaction), the homolytic cleavage of a small number of the  $Si-C_2H_5$  and  $Si-C_3H_7$  bonds takes place and cross-linkages are formed at the positions of rupture.

The structure of the polymers and monomers was established by IR spectroscopy. Thus, the IR spectra



Monomeric Silacyclanes





# Table 2

\*In benzene solution at  $20^{\circ}$ C. \*\*Determined by the light-scattering method. \*\*\*Polymerization was carried out in n-decane solution.

## Table 3

#### Structure at 20°C Thermal oxidation (from x-ray 5 ة temperatureoi analysis)  $Mp, \frac{9}{3}C$  is emperature<br>nitial loss of 5 the first exo<br>sffect,  $\degree$ C Glass point<br>according Poly External  $\frac{n}{100}$  mg/l ပ္စ Structure of the link to Marey,  $mer$ form eight, initial effect, VI  $1,40$  Cryst. 65  $-21.5$ 290  $160 - 175$ White, oqaque, ,<br>сн,сн,сн, solid IX  $3*$  $(C_2H_2)_2$ SiCH<sub>2</sub>CH<sub>2</sub>CH<sub>2</sub>-1.25 <u>.</u> Amorph. Colorless, transparent, rubberiike<br>White, semi-VII 1.35 45  $-54.5$ 260 140 Cryst. сн,сн,сн,– transparent. elastic  $\boldsymbol{\mathrm{X}}$ 200  $3*$  $C_2H_5(C_3H_7)$ SiCH<sub>2</sub>CH<sub>2</sub>CH<sub>2</sub> 1.49 Colorless, trans-Amorph. parent, rubberlike XI  $(CH<sub>3</sub>)$ 2.5 Cryst. 41  $-70$ 200 3203\* .<br>sich.ch.ch White, oqaque,  $\times$ solid

### Properties of the Polymeric Products

<sup>\*</sup>Determined under the polarization microscope and from the DTA curve, \*\*Studied on a derivatograph on the system of F. Paulik, J. Poulik, and L. Erdey, with a rate of heating of  $6^{\circ}C/\text{min}$ , \*\*\*Weakly-<br>expressed diffuse endothermic effects are observed in the 200-300°C region. \*\*\*\*Described previously  $[2].$ 

of the spiran compounds II and III have absorption bands at 930, 1120, and 1185  $cm^{-1}$  which are characteristic for the 1, 1-disubstituted cyclobutane ring.



Fig. 1. X-ray diagram of poly- (trimethylenesilacyelopentane).

In the spectrum of II, absorption bands at 870, 1023, 1077, and 1150  $\text{cm}^{-1}$  characterize the silacyclopentane ring, and in the spectrum of tII, absorption bands at 913 and 982  $\text{cm}^{-1}$  show the presence of the silacyclohexane ring. A study of the IR spectra of the products of the thermal polymerization of II and III has shown that polymerization took place with the opening of the  $Si-C$  bond of the silacyclobutane ring and the formation of linear polymeric products the links of the main chains of which are:

$$
\Big|\overline{\bigcup}_{v_1}^{Si-(CH_2)_3-}\Big|_s \text{ and } \Big|\overline{\bigcup}_{v_1}^{Si-(CH_2)_3-}\Big|_s
$$

respectively. The spectra of VI and VII retain the frequencies of the silacyclopentane and silacyclohexane groups, but the absorption bonds at 930 and  $1120 \text{ cm}^{-1}$ (sflacyclobutane ring) have disappeared, and a whole series of bands characterizing a trimethylene chain between Si atoms has appeared (905, 943, 990, 1025, 1080, and 1140  $\text{cm}^{-1}$ ). The tricyclic compound I be-



Fig. la. X-ray diagram of poly- (trimethyle nesilacyclohexane).

haves similarly. Its IR spectra retains the absorption bands of the silacyclopentane ring (860, 1020, 1030, 1070, and 1150  $\text{cm}^{-1}$ ) these also being observed in the

spectrum of the product of the thermal polymerization of I (VIII). At the same time, the absorption of the disilacyclobutane ring in the monomer at  $940 \text{ cm}^{-1}$ changes into absorption at 1040  $\text{cm}^{-1}$  on passing to the spectrum of the polymer. This shows the opening of the disilaeyclobutane ring and the formation of a polymeric linkage of the type of

VIii

The structures of the 1, 1-dialkylsilacyclobutanes IV and V synthesized and of the heterochain siliconcarbon polymers IX and X obtained from them have been established simiiarly. The properties of the polymers are given in Table 3.

A consideration of the information given in Table 3 may lead to the following conclusions. As suggested, the polymer VII, containing a silacyelohexane framing, possessed a comparatively low glass point  $(-54.5\degree C)$ . Like a number of other polymeric silacyclobutanes [2], because of the high regularity of the structure of the main polymer chain, it contains a considerable volume



Fig. 2. Differential thermal analysis curves.

of crystalline phase (see theX-ray diagram in Fig. la). The polymer with the silacyclopentane framing (VI) crystallizes even morereadilythan VII (Fig. 1) without annealing and melts at a higher temperature; its glass point is correspondingly lower than that of VII. The cause of this difference must apparently be sought in the greater conformational rigidity of the silacyclopentane ring as compared with the silacyclohexane ring just as is the case in their carbon analogs [13]. It is an interesting fact that the polymers with a noncyclic framing of the chain--IX and  $X-$ , which are complete structural analogs of polymers VI and VII, respectively, could not be crystallized at temperatures  $>0$ °C.

The polymer VI proved to be heat-stable. When it was heated in a vacuum of  $10^{-3}$  mm at a rate of  $2^{\circ}$  C/ /min, loss in weight began only at  $330^\circ$  C and active degradation at  $380^\circ$  C; and  $400^\circ$  C the loss in weight was 7%. In the case of the polymer VII the loss in weight under conditions of thermal-oxidative degradation began at  $260^\circ$  C, but strong exothermic effects of oxidation appeared considerably earlier (Fig. 2). In this respect, the polymer with the silacyclohexane framing (VII) differs substantially from the polymer

XI with a dimethyl framing of the main chain, in which weak exothermic effects were observed up to 300°C [2]. At the same time, for xI the loss in weight begins at the same temperature as for VII. The open-chain structural analog (X) begins to decompose considerably earlier, although the exo effects in this are also weakly expressed. It is obvious that the framing silaeyelohexane groups are responsible for this difference, increasing the number of methylene groups for each link of the polymer as compared with X and XI. This is confirmed by the fact that the exothermic effect in the polymer with the smaller number of methylene groups in the framing (Vl) appears at a higher temperature  $(\sim 160^{\circ}$  C). An increase in the number of methylene groups of the framing, which are more active than methyl groups in thermal oxidation, is apparently not the only cause of the strong exo effect. Another possible cause may be the cyclic nature of the polymethylene groups framing the polymer chain, which are bound twice to the silicon atom.

In accordance with existing ideas [14], the thermal oxidation of an organic radical singly bound to silicon  $(Si-R)$  leads to the formation of an  $S<sup>i</sup>-OH$  group and to volatile oxygen-containing products which leave the reaction zone if the polymer is heated in an open system. In the oxidation of the  $CH_2(CH_2)_{n-1}Si$  grouping

**I \_\_J** 

one may expect the formation of oxygen-carbon compounds retaining a bond with the main chain of the

polymer—for example of the type of 
$$
HOSi(CH_2)_{n-1}C \begin{matrix} 0 \\ H^{\dagger} \end{matrix}
$$

**and** therefore remaining within the polymer mass.

Aldehyde groups fixed in this way may actively promote the development of chain oxidation processes in the mass of the polymer and cause exothermie effects.

### EXPERIMENTAL

1,1-Tetramethylene-l-silacyclobutane (II). The Grignard reagent prepared from 216 g of 1, 4-dibromobutane and 72 g of magnesium turnings was added rapidly (with ice cooling) to an ethereal soiution of 95 g of 1, 1-dichloro-1-silacyelobutane. The organic layer was separated off and distilled in vacuum, giving 35 g (49%) of II. The physical constants are given in Table 1.

1, l-Pentamethylene-l-silacyolobutane (III). With cooling, 57 g of 1, 1-dichloro-l-silacyelobutane was rapidly added to the Grignard reagent obtained from 92 g of 1, 5-dibromopentane and 36 g of magnesium turnings in ether. A working-up process similar to that of the preceding experiment yielded 19 g (34%) of III, the constants of which are given in Table 1.

1, i:3, 3-Bis(tetramethylene)-l, 3-disilaeyelobutane (I). One hundred grams of magnesium turnings activated with iodine was added to 25 g of 1-chloro-1-chloromethyl-l-silacyclopentane in ether. The reaction mixture was heated for 10 hr and filtered. The yield of I was 6.1 g  $(21\%)$  (see Table 1).

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